

TEM observation of nonamphiphilic copolymer micelles

Eri Yoshida

Received: 21 August 2006 / Accepted: 22 January 2007 / Published online: 1 March 2007
© Springer-Verlag 2007

Abstract Light scattering and transmission electron microscope (TEM) measurements were performed for micelles of a nonamphiphilic poly(vinylphenol)-*block*-polystyrene diblock copolymer (PVPh-*b*-PSt) to determine the shape of the micelles. The micelles were prepared by the self-assembly of the copolymer in 1,4-dioxane, a nonselective solvent, in the presence of 1,4-butanediamine. The logarithm of the normalized time correlation function of the scattered field, $\ln G_1(\tau)$, linearly decayed versus the delay time, τ . The diffusion coefficient measured in the range of the scattering angles from 30° to 150° was almost independent of the square of the magnitude of the scattering vector. The linear decay of $\ln G_1(\tau)$ vs τ and the angular-independence of the diffusion coefficient suggested that the monodisperse spherical micelles were formed by the micellization. The TEM observations confirmed the formation of uniform spheres.

Keywords Poly(vinylphenol)-*block*-polystyrene · A nonamphiphilic block copolymer · 1,4-Butanediamine · Monodisperse spherical micelles · Light scattering · TEM observation

Introduction

Micelles are nanoscale supramolecules formed not only naturally, but also artificially, by molecular self-assembly through noncovalent bonding such as van der Waals interaction, electrostatic attraction, and hydrogen bonding.

The micelles can solubilize and stabilize substances by taking them into the micellar cores. The micelles have many applications as dispersing agents and stabilizers in industry. Examples include carriers in drug delivery [1, 2], stabilizers in food processing [3, 4], dispersing agents for coal–water mixture fuels [5] and for coatings of dyeing and painting [6], and surfactants in water purification [7] and in extraction technologies [8]. Recently, many publications have been released on the micelle formation of block copolymers. The micelle formation concerns not only the direct micellization of amphiphilic copolymers [9–11], but also the indirect micellization of nonamphiphilic copolymers consisting entirely of solvophilic blocks. The indirect micellization is induced by variation in the external parameters such as temperature [12, 13], pressure [14–16], pH [17, 18], and salt formation [19, 20]. During the micellization, molecularly dissolved nonamphiphilic copolymers are converted in situ to amphiphilic copolymers by these stimuli. Hence, the indirect micellization has many advantages over the direct micellization. There is no dependence on the solubility balance of the solvophilic and solvophobic moieties when designing the nonamphiphilic copolymers. It is able to provide a better selection of the driving force. In addition, a variety of amphiphilic copolymers can be created from one nonamphiphilic copolymer in situ by selecting the stimuli.

I have recently observed the micelle formation of nonamphiphilic copolymers using the interaction with additives [21–27]. The poly(vinylphenol)-*block*-polystyrene diblock copolymer (PVPh-*b*-PSt) is self-assembled into micelles in nonselective solvents in the presence of α , ω -diamine [28–32]. The micelles were formed by the hydrogen bond cross-linking between the PVPh blocks via the diamine. Due to the unusual micelle formation, the shape of the micelles has been attracting increasing

E. Yoshida (✉)
Department of Materials Science,
Toyohashi University of Technology,
1-1 Hibarigaoka, Tempaku-cho,
Toyohashi, Aichi 441-8580, Japan
e-mail: eyoshida@tutms.tut.ac.jp

attention. I revealed that the PVPh-*b*-PSt copolymer formed spherical micelles with the diamine based on a light scattering study and transmission electron microscopy (TEM) observations. This article describes the light scattering studies and TEM observations of the micelles.

Experimental

Instrumentation Light scattering measurements were performed by a Photol Otsuka Electronics ELS-8000 electrophoretic light scattering spectrophotometer equipped with a system controller, an ELS controller, and a He–Ne laser operating at $\lambda=632.8$ nm and also by a Photol Otsuka Electronics DLS-7000 super dynamic light scattering spectrometer equipped with LS-71 control unit, an LS-72 pump controller, and an argon ion laser operating at $\lambda=488$ nm. TEM measurements were performed with a JEOL JEM-2010 electron microscope.

Materials PVPh-*b*-PSt was prepared as reported previously [31]. The molecular weight of the copolymer was M_n (PVPh-*b*-PSt) = 10,000-*b*-120,000 by ^1H NMR. 1,4-Butanediamine (BDA) was distilled over calcium hydride. 1,4-Dioxane was purified by refluxing on sodium for several hours and distilled over sodium.

Light scattering measurements PVPh-*b*-PSt (10 mg) was dissolved in 1,4-dioxane (3 ml) and, using a syringe, the resulting solution was injected through a microporous filter into a cell. The solution was subjected to light scattering measurements at 10 °C. After the measurement, 24 μl of BDA solution [BDA (148 mg, 1.68 mmol) in 1 ml of 1,4-dioxane] was added to the copolymer solution in the cell, and the mixture was vigorously shaken. The solution was allowed to stand at 10 °C for 5 min and then subjected to light scattering again.

TEM observation A drop of the micellar solution was allowed to fall on a Cu grid with a carbon substrate, and the solvent was immediately evacuated using a filter paper. The grid was dried in air for a few hours and then subjected to TEM measurements.

Results and discussion

The time correlation function of the scattered intensity is given by Eq. 1

$$G_2(\tau) = A(1 + \beta|G_1(\tau)|^2) \quad (1)$$

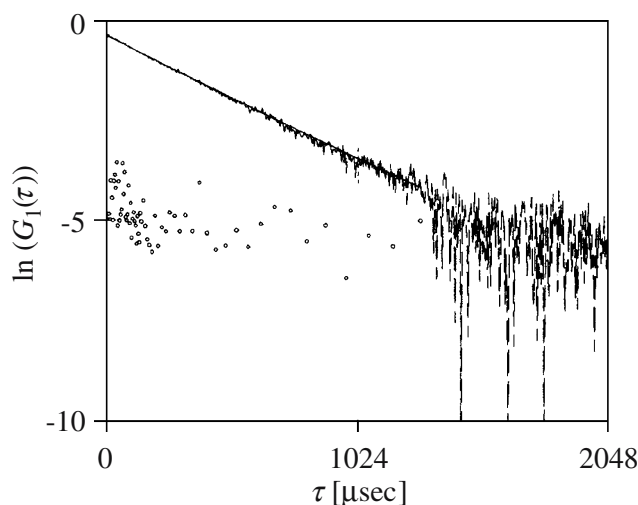


Fig. 1 The plots of $\ln G_1(\tau)$ vs τ for the PVPh-*b*-PSt micelles prepared at BDA/VPh of 6 at 10 °C in 1,4-dioxane at a copolymer concentration of 3.33 mg/ml. The scattering angle $\theta=90^\circ$

where A is a baseline that corresponds to the square of the average scattering intensity, β is an efficiency factor, and $G_1(\tau)$ is the normalized time correlation function of the scattered field. For spherical particles with a monodispersity, $G_1(\tau)$ displays a single exponential decay curve and is represented by Eqs. 2 and 3

$$G_1(\tau) = \exp(-\Gamma\tau) \quad (2)$$

$$\ln(G_1(\tau)) = -\Gamma\tau \quad (3)$$

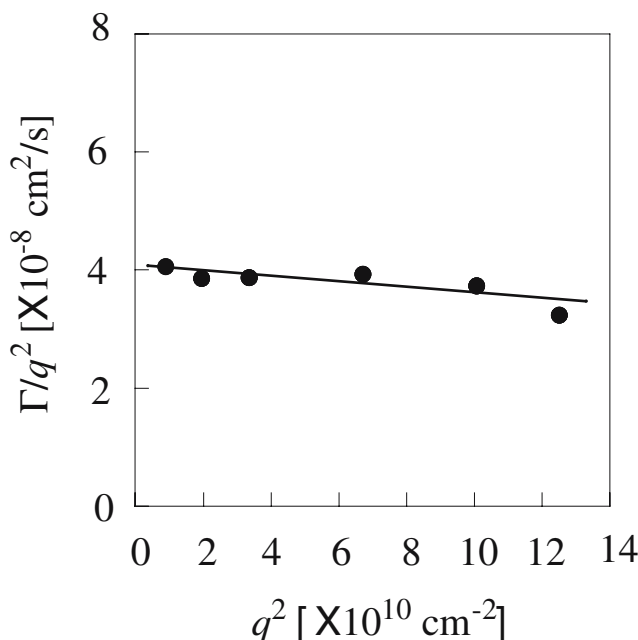
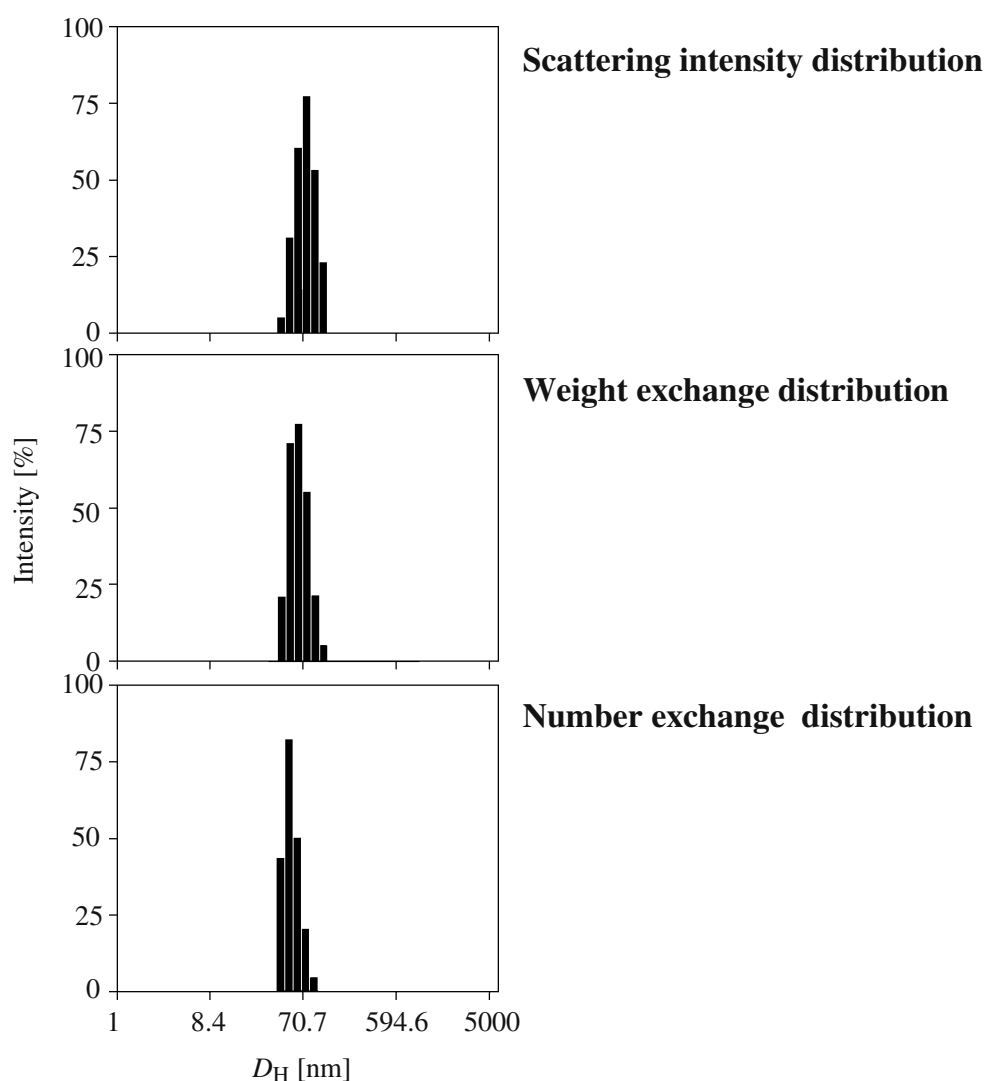


Fig. 2 The plots of D vs q^2 for the PVPh-*b*-PSt micelles prepared at BDA/VPh of 3 at 10 °C in 1,4-dioxane at a copolymer concentration of 3.33 mg/ml

Fig. 3 The diameter distributions obtained by the Marquadt analysis of the PVPh-*b*-PSt micelles prepared at BDA/VPh of 6 in 1,4-dioxane at 10 °C at a copolymer concentration of 3.33 mg/ml. $\theta=90^\circ$



where Γ is a decay constant and τ is a delay time [33]. The PVPh-*b*-PSt diblock copolymer produced micelles at 10 °C in 1,4-dioxane in the presence of BDA over a molar ratio of 3 for the BDA the VPh unit (BDA/VPh) [31]. The hydrodynamic diameter of the micelles was unchanged over a BDA/VPh of 3; however, the micelles prepared at a BDA/VPh of 6 were more thermally stable than those at 3. Figure 1 shows the plots of the $\ln G_1(\tau)$ versus τ for the micelles prepared at the BDA/VPh of 6. The $\ln G_1(\tau)$ linearly decayed vs τ , suggesting the formation of nearly monodispersed spherical micelles.

The analysis of the angular dependence of the micelles also supported the formation of nanospheres with a narrow size distribution. The decay constant Γ is represented as Eqs. 4 and 5

$$\Gamma = q^2 D \quad (4)$$

$$q = (4\pi n_0 / \lambda_0) \sin(\theta/2) \quad (5)$$

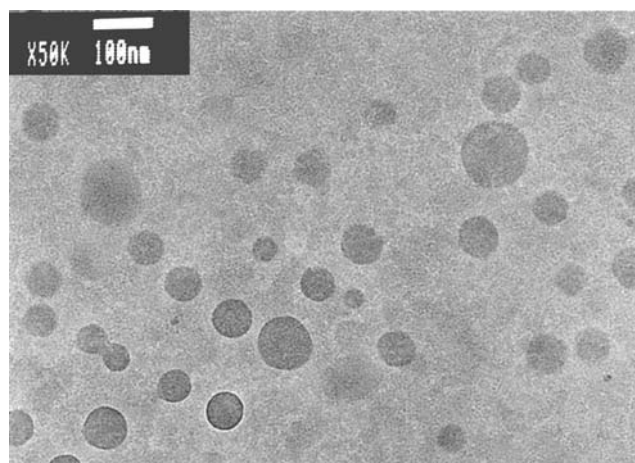


Fig. 4 The TEM image of the PVPh-*b*-PSt micelles prepared at BDA/VPh of 6 at 10 °C in 1,4-dioxane at a copolymer concentration of 3.33 mg/ml

where q is the magnitude of the scattering vector, D is the diffusion coefficient, n_0 is the solvent refractive index, λ_0 is the wavelength of the laser, and θ is the scattering angle. The diffusion coefficient, $D = I/q^2$ measured in the scattering angle range from 30° to 150° was almost independent of q^2 for the micelles (Fig. 2). This angular-independence of the diffusion coefficient was also observed for the micelles prepared in other nonselective solvents such as THF and ethyl acetate, although the amount of BDA required for the micellization varied with the solvent. It is suggested that PVPh-*b*-PSt produces spherical micelles with a narrow size distribution in these solvents in the presence of BDA.

The hydrodynamic diameter of the micelles prepared at the BDA/VPh of 6 in 1,4-dioxane was estimated to be 64.3 nm by the cumulant analysis, whereas the aggregation number was 15 based on the relative scattering intensity [15, 29]. Figure 3 shows the following three different distributions obtained by the Marquadt analysis [34] of the micelles: scattering intensity distribution, weight exchange distribution, and number exchange distribution. The hydrodynamic diameters of the micelles were estimated to be $D_H(\text{Is}) = 70.5 \pm 16.5$ nm by the scattering intensity distribution, $D_H(\text{wt}) = 59.8 \pm 14.1$ nm by the weight exchange distribution, and $D_H(\text{no}) = 51.8 \pm 10.5$ nm by the number exchange distribution. The distribution index of the hydrodynamic diameter was $D_H(\text{wt})/D_H(\text{no}) = 1.15$ based on the weight exchange and number exchange distributions.

The TEM observations revealed that spherical micelles were formed by the self-assembly of PVPh-*b*-PSt by BDA. The TEM image of the micelles is shown in Fig. 4. Almost complete spheres were observed for the PVPh-*b*-PSt micelles. The weight average diameter of the micelles was estimated to be $D(\text{wt}) = 78.5$ nm, whereas the number average diameter was as $D(\text{no}) = 65.0$ nm based on the TEM measurements. The distribution index of the diameter is given by the following equation [35]

$$D_{(\text{wt})}/D_{(\text{no})} = \{\sum N_i D_i^4 / (\sum N_i D_i^3)\} / \{\sum N_i D_i / \sum N_i\} \quad (6)$$

where N_i is the number of particles with diameter D_i . The size distribution was estimated to be $D(\text{wt})/D(\text{no}) = 1.21$. There is a small difference in the size distributions estimated by the TEM and light scattering methods. The number average diameter was in good agreement with that determined by the cumulant analysis, indicating that the micelles in the solution maintained their size in air. In some cases, amphiphilic copolymer micelles showed a smaller size in the TEM than that in light scattering due to swelling of the micellar cores in solution. The cores of the PVPh-*b*-PSt micelles consist of the PVPh blocks cross-linked by BDA. Therefore, the micellar size should be independent of the presence of the solvent molecules in the cores.

Conclusions

The light scattering and TEM measurements confirmed that the PVPh-*b*-PSt and BDA produced spherical micelles with a narrow size distribution. The micellar size estimated by the cumulant analysis during the light scattering was in good agreement with that by the TEM observations. This negligible difference in size implied that the micelles in the solution maintained their size in air on the basis of the cross-linked cores of the PVPh blocks by BDA. There was also a small difference in the size distribution between the light scattering and TEM observations. This is the first study demonstrating that the nonamphiphilic block copolymer formed spherical micelles by the interaction with the additive. This micelle formation is expected to be a new method for preparing nanospheres with a narrow size distribution.

Acknowledgment We thank Mr. Hirokazu Muramoto for his aid with the TEM characterization. We are also thankful for the JSPS Grant-in-Aid for Science Research #16510086, the Amano Engineering and Technology Center, and the grant for young researchers project of the Research Center for Future Technology, Toyohashi University of Technology for the financial support of this study.

References

1. Yoon KA, Burgess DJ (1997) *J Pharm Pharmacol* 49:478
2. Lawrence MJ, Lawrence SM, Barlow DJ (1997) *J Pharm Pharmacol* 49:594
3. Lin JJ, Knifton JF (1991) *J Organomet Chem* 417:99
4. Dralle-Voss G, Oetter G, Wekel H, Oppenlaender K (1996) *Ger Offen DE*, 19505100 A1 (22 Aug)
5. Ogura T, Tanoura M, Hiraki A (1993) *Bull Chem Soc Jpn* 66:1343
6. Cohen D, Hitchcock E, Pohl S (1995) *Eur Pat Appl*, EP 640335 A2 (1 Mar)
7. Park JW, Jaffe PR (1993) *Environ Sci Technol* 27:2559
8. Ismael M, Tondre C (1992) *Langmuir* 8:1039
9. Patrickios CS, Forder C, Armes S, Steven P, Billingham NC (1996) *J Polym Sci A Polym Chem* 34:1529
10. Allgaier J, Poppe A, Willner L, Richter D (1997) *Macromolecules* 30:1582
11. Resendes R, Massey J, Dorn H, Winnik MA, Manners I (2000) *Macromolecules* 33:8
12. Lowe AB, Billingham NC, Armes SP (1997) *Chem Commun* 1035
13. Arotcarena M, Heise B, Ishaya S, Laschewsky A (2002) *J Am Chem Soc* 124:3787
14. McClain JB, Canelas DA, Samulski ET, DeSimone JM, Londono JD, Cochran HD, Wignall GD, Chillura-Martino GD, Triolo R (1996) *Science* 274:2049
15. Buhler E, Dobrynin AV, DeSimone JM, Rubinstein M (1998) *Macromolecules* 31:7347
16. Zhou S, Chu B (1998) *Macromolecules* 31:5300
17. Liu S, Weaver JVM, Tang Y, Billingham NC, Armes SP (2002) *Macromolecules* 35:6121
18. Hu Y, Kramer MC, Boudreaux CJ, McCormick CL (1995) *Macromolecules* 28:7100

19. Liu S, Zhang G, Jiang M (1999) *Polymer* 40:5449
20. Brontein LM, Sidorov SN, Valetsky PM (1999) *Langmuir* 15:6256
21. Yoshida E, Tanaka M, Takata T (2005) *Colloid Polym Sci* 283:1100
22. Yoshida E, Tanaka M, Takata T (2005) *Colloid Polym Sci* 284:51
23. Yoshida E, Ohta M (2006) *Colloid Polym Sci* 284:718
24. Yoshida E, Ohta M (2005) *Des Monomers Polym* 8(5):501
25. Yoshida E, Ohta M (2005) *Colloid Polym Sci* 283:872
26. Yoshida E, Hironaka A (2004) *Polym J* 36:248
27. Yoshida E (2003) *Polym J* 35:484
28. Yoshida E, Kunugi S (2002) *J Polym Sci A Polym Chem Ed* 40:3063
29. Yoshida E, Kunugi S (2002) *Macromolecules* 35:6665
30. Yoshida E (2003) *Polym J* 35:965
31. Yoshida E, Ohta M, Terada Y (2005) *Polym Adv Technol* 16:183
32. Yoshida E, Itsuno S (2005) *Colloid Polym Sci* 284:19
33. Brown W (1996) *Light scattering principles and development*. Clarendon, Oxford, pp 439–442
34. Marquardt DW (1963) *J Soc Ind Appl Math* 11:431
35. Kobayashi S, Uyama H, Yamamoto I, Matsumoto Y (1990) *Polym J* 22:759